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Doppler-broadening measurements of positron annihilation with high-momentum electrons in pure elements

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Abstract

Doppler-broadening measurements of the electron–positron annihilation line in twenty-seven single-element samples are presented. A coincidence technique has been used to suppress the background and to evidence the contribution of positron annihilation with core electrons. Systematic dependences on the atomic number of the target material are found in ratio curves obtained dividing the measured spectra by the spectrum of a reference material. The positron lifetime technique has been used to detect the presence of positron traps in all the samples. The change in the high-momentum part of the annihilation line due to positron trapping is illustrated. The measured data are in a good qualitative agreement with recent theoretical calculation and constitute the most complete measurement series, up to now, to establish a future data-base for positron annihilation spectroscopy.

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1. Introduction

Positron annihilation spectroscopy (PAS) is a well-established technique to detect open volume and negatively charged defects in solids [1]. Positrons, injected in a solid, lose their energy in a few picoseconds and, after a diffusion motion, annihilate mostly in two 511 keV gamma quanta with an electron of the medium. Due to their positive

charge, positrons are efficiently trapped by open volume defects.

The two most frequently used techniques to detect and to investigate the properties of open volume defects and their interaction with the medium are the lifetime and the Doppler broadening techniques. If fast positrons from a radioactive nuclide are used, the mean properties of the material under investigation are measured in a layer several tens of micron thick [2]. Otherwise, if a slow positron beam is utilized, a depth scanning from the surface to a few microns is possible [1].

In the Doppler broadening technique the 511 keV annihilation line, broadened by the Doppler

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shift due to the total momentum of the positron–electron annihilation pair, is directly recorded. In the measured spectrum, which is the one-dimensional momentum distribution of the annihilating positron–electron pair, two main regions can be distinguished: a low-momentum region close to the peak position (typically $0 < p_L < 3.0 \times 10^{-3} m_0 c$, where p_L is the longitudinal momentum) which reflects annihilations with valence electrons and a high-momentum region toward the tails of the spectrum (typically $6.2 \times 10^{-3} m_0 c < p_L < 15 \times 10^{-3} m_0 c$) due to annihilation with high-momentum electrons. Usually two integral parameters (S and W) are used to probe the two regions of the spectrum [3]. With these two parameters it is possible to extract information on the type and concentration of the open volume defects [4]. On the other hand, the shape of the 511 keV annihilation line is different for each element. In particular the core electrons maintain their atomic characteristic also in a solid making it possible to use the signal from the positron–core electron annihilation for chemically selecting the atoms. A drawback is that the positron annihilation probability with electrons of high momentum is very low and the amplitude of the signal is comparable with the background. As a consequence, a background reduction coincidence technique with two detectors must be used to extract information from the tails (p_L up to $50 \times 10^{-3} m_0 c$) of the spectrum [5,6] and this requires an increase of the measurement time for each spectrum. With a more detailed analysis of the Doppler spectrum one can get more and unique information about the defect site. The wave function of a trapped positron overlaps with the atoms surrounding the defect and these atoms can be identified by analyzing the high-momentum part of the annihilation spectrum.

The possibility to identify not only the type of open volume defect, but also the atoms that surround the defect, makes the positron probe even more powerful. Recently the promising problem of analyzing the positron annihilation with core electrons has been attacked both experimentally and theoretically [7–25]. The chemical selectivity of the positron probe can be used to detect impurity atoms associated to open volume defects. Refs. [7,8,10,22,23,25] have demonstrated the capability

of the technique to identify the detailed atomic structure of vacancy–impurity complexes in semiconductors.

In order to identify atoms that surround an open volume defect, we need to know accurately the core annihilation signal from these atoms and to describe it by theoretical models. Until now, measurements and calculation [9,12,24] of positron–core-electron momentum were done on some pure elements but the picture is far from being complete. Discrepancies exist between different sets of measured and calculated data [9,24]. To evidence the small differences of positron annihilation with core electrons of different materials we shall present the experimental momentum spectra reduced to “ratio-curves” obtained dividing the measured curves by the curve of a reference element [9]. In this work we have measured positron annihilation spectra for 27 pure elements and we will outline regularities in the position and amplitude of peaks in the ratio curves. The pure elements have been characterized from a positronic point of view (i.e. absence or presence of open volume defects) with lifetime measurements. The lifetime of a positron in the solid is inversely proportional to the electron density probed by the positron itself. In open volume defects the electron density is reduced and the lifetime of a trapped positron increases in comparison with the lifetime of a free positron annihilating in the bulk of the material. Defects with different open volumes can be distinguished because trapped positrons show different lifetimes. The change of the ratio curves due to the presence of defects will be discussed, showing how the ratio curves are very sensitive to a small amount of positron traps. Our results will also be compared, when possible, with recent theoretical calculations [24] and measurements [9].

2. Experimental

Twenty-seven pure elements have been investigated; some of them were in single crystalline form. The list of these elements and some characteristics of the measured samples are reported in Table 1. The chemical purity of the samples is given in column two.

Table 1

Atomic number Z , measured elements, purity of the elements, temperature used for the thermal treatments (T_a), average positron lifetimes τ_a

Z	Elements	Purity (%)	T_a °C	τ_a (ps)	τ_1 (ps)	τ_2 (ps)	I_2 (%)	τ (ps)
4	Be	–	1000	156	138 ± 1	298 ± 7	12.2 ± 1	137/142/213
5	B	99.5	1100	209	–	–	–	–/–/–
6	C (graphite)	99.5	1100	264	170 ± 3	343 ± 3	54.6 ± 1	–/215/–
12	Mg	–	250	224	–	–	–	237/225/230
13	Al (111)	99.9	500	160	–	–	–	166/163/166
14	Si p(100)	–	–	220	–	–	–	221/–/228
22	Ti	99.9	1100	145	–	–	–	146/147/148
23	V	99.7	1100	123	–	–	–	116/130/125
24	Cr	99.5	1100	157	83.1 ± 1	248 ± 2	44.8 ± 1	101/120/–
25	Mn	99.5	–	150	121 ± 1	279 ± 6	17.7 ± 1	103/–/–
26	Fe	99.99	1000	123	110.7 ± 1	257 ± 10	8.6 ± 1	101/106/107
27	Co	99.95	1100	117	99.6 ± 1	245 ± 10	11.7 ± 1	97/–/119
28	Ni (111)	99.98	1000	105	–	–	–	96/110/110
29	Cu (100)	99.8	850	115	–	–	–	106/110/122
30	Zn (0001)	99.9	250	149	–	–	–	134/148/160
31	Ga	99.9	–	237	–	–	–	–/190/194
32	Ge (100)	–	–	227	–	–	–	228/230/234
40	Zr	99.8	1100	159	–	–	–	159/165/165
41	Nb	99.8	1100	127	–	–	–	122/119/122
42	Mo (100)	99.9	>1500	119	–	–	–	111/103/121
47	Ag	99.9	850	135	–	–	–	120/131/138
50	Sn	99.9985	150	197	–	–	–	275/289/201
72	Hf	99.5	>1500	179	–	–	–	149/–/–
74	W	99.8	>2000	115	103 ± 1	244 ± 16	8.6 ± 1	100/105/120
78	Pt	99.5	1000	167	142 ± 1	282 ± 8	18 ± 1	94/99/117
79	Au	99.99	850	123	119 ± 2	232 ± 56	3.7 ± 1	107/117/118
82	Pb	99.5	250	206	–	–	–	187/194/214

The error on the average lifetimes is 1 ps. First (τ_1), second lifetime (τ_2) and the intensity I_2 of the second lifetime are reported for those elements in which the average lifetime was different from the bulk lifetime (columns from six to eight). In the last column collection of bulk lifetime τ from literature: first value (theoretical) from [30], second and third value (experimental) from [31,32] respectively.

The positron lifetime and the Doppler-broadened 511 keV annihilation line have been measured after annealing of the samples. In some cases measurements were also done on as-received samples. In general, the as-received samples can show presence of several types of positron traps: open volume defects from vacancies to vacancy clusters, dislocations, voids. All the samples were annealed for three hours in a vacuum of about 5×10^{-7} mbar (annealing temperatures are given in third column of Table 1). Some of the samples (Cr, Co, Au) have been annealed in two or more cycles of three hours. Mo, Hf, W were annealed at high temperatures in vacuum by electron bombardment. The samples were cooled down from the annealing temperature to room temperature in

the oven by turning off the heating. Normally the entire cycle took more than 12 h.

For lifetime and Doppler broadening measurements a ^{22}Na positron source was sandwiched between two equal samples. The ^{22}Na positron sources (0.37–0.6 MBq, prepared by deposition of $^{22}\text{NaCl}$) were sandwiched between two kapton foils 7 μm thick.

The 511 keV annihilation line is Doppler-broadened ($511 \pm \Delta E$) due to the longitudinal momentum p_L component of the annihilating positron–electron pair. In a Doppler experiment the longitudinal component p_L in the direction of the detector is measured. The momentum p_L is correlated to the Doppler shift ΔE by the formula $p_L = 2\Delta E/c$ where c is the light velocity. To study

the annihilation with core electrons, the background must be reduced using a two-detector coincidence system [5]. In this experiment the main detector was a high purity Ge (HPGe) detector (16% efficiency) with a resolution of 1.3 keV at 511 keV with 2 μ s shaping time in the HPGe spectroscopy amplifier. A pileup rejection was used. The auxiliary detector supplying the coincidence signal was a NaI(Tl) scintillator. It was placed in collinear geometry with the Ge detector. The two detectors were shielded with lead in order to decrease the noise. The distance of the samples from the detectors was 23 cm. With this set up, we get a coincidence count rate of about 140 counts/s in the 511 keV peak and a peak to background ratio of 1.5×10^4 on the high-energy side of the peak (see Fig. 1). The coincidence spectra were taken until more than 1×10^7 counts had been accumulated in the peak. We recall that using a NaI(Tl) detector coupled with the HPGe detector, only the background from the high-energy part of the 511 keV line can be removed. The effects due to the incomplete charge collection, the Compton scattering and 3-gamma positronium decay cannot be eliminated in the low energy side of the peak. After each spectrum was recorded, the energy scale of the system was calibrated by taking the peak position

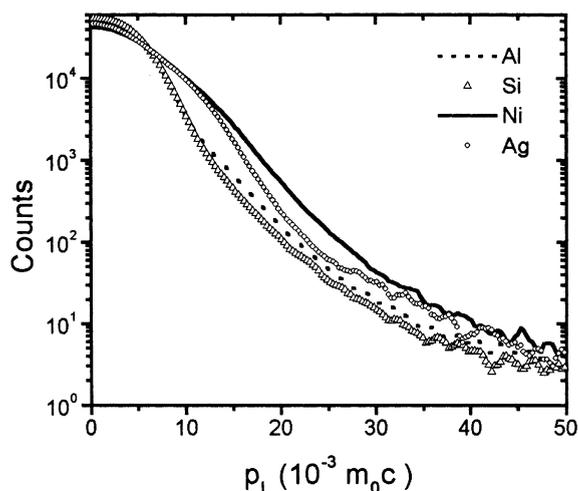


Fig. 1. Doppler spectra of Al, Si, Ni and Ag obtained by the coincidence technique. The curves, corrected for the background and the source component, are normalized to the same total area.

of the 356.23 keV gamma line supplied by a ^{133}Ba source in the single detector mode. In these spectra, each channel was 77.6 eV wide.

Lifetime measurements have been done to check if the samples were defect free, as seen by positrons. The emission of a positron from a ^{22}Na radioactive source is followed by a 1.28 MeV de-excitation photon. In a lifetime spectrometer this photon is used as a start signal and the 511 keV annihilation photon as a stop signal. The positron lifetime measurements presented in this work, have been carried out with a fast-fast coincidence system [26] with Hamamatsu R2083Q photomultipliers and BaF_2 scintillators. The resolution of the apparatus was 160 ps (FWHM) when measuring the positron lifetime. More than 1×10^6 counts have been accumulated in every lifetime spectrum. The positron lifetime is related to the delay between the positron emission and its annihilation. A lifetime spectrum is the sum of exponential decay components with different intensities. In a pure element without positron trapping centres, after correction for the source component, the lifetime spectrum is composed of a single exponential that gives the characteristic free positron annihilation lifetime in the bulk of the system τ [3]. In the following we have characterized our lifetime spectra by the average lifetime τ_a . Our τ_a is evaluated with an error of about 1 ps. In the case of a pure element without defects, τ_a equals τ (only one lifetime is present in the spectrum). If open volume defect are present, τ_a is larger than τ of the element without defects.

3. Analysis of the data

First we consider positron lifetimes. The lifetime spectra have been analyzed using the Positronfit program [27]. To extract the positron lifetime in the sample, the fraction of positrons annihilating in the source and in the source support (source correction) must be subtracted from every spectrum. Kapton, here used as supporting material for the radioactive source, has a very simple lifetime spectrum with a single lifetime of 384 ps. In kapton no positronium is formed [28]. Because of the fraction of backscattered positrons from the

samples changes with the atomic number Z of the material, the source correction was done by subtracting the kapton lifetime with a different intensity according to the empirical formula given in [3,29]. The intensity of positron annihilating in the kapton foil spans from 4% for the lightest elements to 20% for the heaviest elements. A long lifetime component (1.5 ns) with a fixed intensity of 0.12%, arising from positrons annihilating at the surfaces, was also subtracted from every spectrum. The average lifetime τ_a measured after the last annealing are reported in column five of Table 1. By comparison, values from literature [30–32] are listed in column nine. To our knowledge the lifetime data in [30–32] are the only collection published until now. Note that all our lifetime data reported in Table 1 have been measured with the same set up and analyzed with the same procedure.

After repeated annealing some samples have shown residual positron trapping, their average lifetime are reported in bold characters in Table 1. For these samples we have also reported the two lifetimes (τ_1 and τ_2), in which the spectrum was deconvoluted and the intensity (I_2) of the second lifetime (columns from six to eight). For all these samples the second lifetime ranges from 240 to 300 ps and can be attributed to residual small open volume defects, most probably vacancy clusters formed for vacancy migration during heating [3].

The change in shape of the Doppler broadened spectrum due to annihilation with core electrons is small and, in many cases, it is difficult to appreciate directly the differences from one element to the other. As an example some selected spectra for pure elements are shown in Fig. 1. All the 27 measured curves approximately lie between the Si and the Ni curves.

Nevertheless, in order to enhance the differences among different spectra, it is common practice to construct ratio curves (10, 23, 25). The spectra under examination are divided by the spectrum of a material chosen as the reference. The choice of the reference material determines the shape of the ratio curves: therefore the reference has to be chosen carefully. To give an example, investigating a series of iron alloys, a good choice could be to have a pure iron reference. In our case, willing to make a systematic work on many elements, the

choice has been dictated by three criteria: first, the reference material has to be the same for all samples; second it has to be a low Z element; third it has to be among those already used by previous authors – in order to allow the comparison with previous results. The use of a single low Z reference element for all samples can make easier the comparison with theoretical results (9, 24). The spectra shown in Fig. 2 and following have been obtained as a ratio to the aluminum spectrum.

Starting with the element spectrum a constant background calculated as the average of the counts in 100 channels from 607 to 615 keV was subtracted. Then a source correction was applied to every spectrum. We have prepared and measured two thick kapton samples. The Doppler spectrum of kapton was subtracted from all the measured spectra with a percentage weight dependent on atomic number of the sample and equal to the one used in the evaluation of the average lifetime reported in Table 1. Without this correction, errors up to -15% in the height of the peaks of the ratio curves can be made in the heavier elements. Also the relative height of the peaks, where more than one peak is present, changes. Finally a smoothing routine has been applied (averaging over n channels, with n increasing from 9 to 16, going from 511 to 530 keV) to the corrected data and the spectra have been normalized to a total area of 1.0 from 511 to 530 keV (p_L from 0 to $74.3 \times 10^{-3} m_0 c$). This smoothing procedure is necessary to extract the behaviour of the high-momentum tail of the spectrum, because the signal to noise on the counts in each channel decreases as the energy of the annihilation gamma increases (for a discussion see [10]). In our spectra an error of 3% is reached at about 516 keV (p_L around $23 \times 10^{-3} m_0 c$) and 12% at about 524 keV (p_L around $50 \times 10^{-3} m_0 c$). The corrected and normalized spectra have been finally divided by the reference spectrum.

In order to obtain a reference spectrum we have chosen two single crystal Al (111) samples. As stressed in [21] it is important to record the orientation of the reference material because the ideal Doppler profile of Al contains a kink at the Fermi momentum that can affect the ratio curves. On the other hand, Doppler measurements reported on the same references on Al samples with two different

orientation do not show any appreciable difference. To minimize the noise on the ratio curves, particular care was taken to prepare the reference curve. First, a long measurement was performed accumulating about 3×10^7 counts. Then, after background subtraction and source correction, a nine points smoothing routine was applied and finally the long tail from 518 keV was substituted by a two exponential fitting curve. This last step is justified by the fact that the Doppler broadening spectrum of Al, as can be seen from theoretical calculation [7,24], decreases monotonically in the high-momentum region.

4. Results and discussion

4.1. Pure elements

The detailed shape of the high-momentum part of the Doppler spectra depends on the different contributions coming from the positron annihilation with the electrons of each shell of the atom. Positrons annihilate mainly with the electrons of the outermost shell due to the repulsion of the nucleus. Among the factors that determine the increase of the positron–electron annihilation probability at certain momenta we can remember: (i) the increase of the number of electrons at the individual shells; (ii) the localization in the r space of the electrons; more localized electrons gives a broad momentum distribution and vice versa; (iii) the crystalline structure. Calculations like in [7–9,24] will be needed for each element to set the partial contribution of positron–electron annihilation at the measured total momentum distribution and to the peaks appearing in the ratio curves. On the other hand phenomenological considerations can be done observing different groups of ratio curves and extrapolating the main results, obtained for some elements, in the above cited references.

We have grouped our Doppler broadening measurements on pure elements according to their position in the Periodic Table and to some similarities observed in their ratio-curve. The groups are the following: Be, B, C, Mg (Fig. 2), Ti, V, Cr, Mn, Fe, Co, Ni, Cu, Zn, Ga (Fig. 3(a) and (b)), Zr,

Nb, Mo, Ag (Fig. 4(a)), Hf, W, Pt, Au (Fig. 4(b)), Si, Ge, Sn, Pb (Fig. 5).

The Be, B, C curves in Fig. 2 show a single peak followed by a valley around $19.5 \times 10^{-3} m_0 c$. For these light elements no theoretical calculations have been published. The peaks around $10 \times 10^{-3} m_0 c$ of Be, B and C can be correlated to the positron annihilation with the outermost electrons 2p. The peak rise with the number of 2p electrons. Mg (with the 3s orbital filled) is the element in which positrons have the highest probability to annihilate with low-momentum electrons. This property reflects in a particular ratio curve with a maximum at zero momentum followed by a valley at $6 \times 10^{-3} m_0 c$ and two peaks of decreasing intensity at 12.2×10^{-3} and $20 \times 10^{-3} m_0 c$.

In Fig. 3 (note the different scale for the vertical axis in Fig. 3(a) and (b)) the ratio curves for the transition metals of the fourth row in the periodic table are reported, together with the Ga data. The elements of this series are characterized by a well-defined peak slightly above $10 \times 10^{-3} m_0 c$, with increasing intensity going from Ti to Cu. Starting from vanadium an other shoulder manifests around $20 \times 10^{-3} m_0 c$. This shoulder increases in amplitude and, going from Fe to Cu through Co and Ni, broadens the ratio curves towards higher

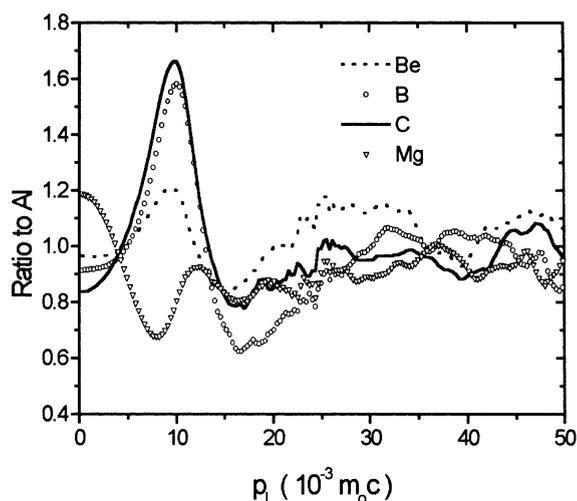


Fig. 2. Ratio of the experimental Be, B, C, Mg, positron–electron momentum distribution to the positron–electron momentum distribution of Al.

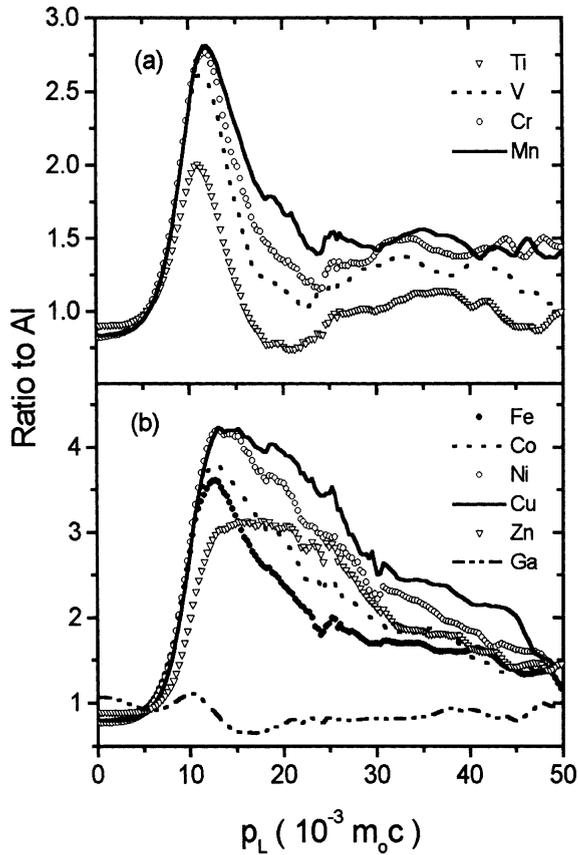


Fig. 3. Ratio of the experimental (a) Ti, V, Cr, Mn and (b) Fe, Co, Ni, Cu, Zn, Ga, positron–electron momentum distribution to the positron–electron momentum distribution of Al.

momenta. The Ni and Cu ratio curves have the same value in the maximum and differentiate in the $(15 < p_L < 30) \times 10^{-3} m_0 c$ range where the Cu curve is higher. Also the peak position of the elements of this series moves progressively towards higher momenta. Calculations of positron–electron momentum distribution on Cu [8,24], Ni and Fe [24] have shown that in the $(15 < p_L < 40) \times 10^{-3} m_0 c$ range the dominant contribution is due to positron annihilation with 3d electrons. On the other hand, in Cu the 3p electrons give the dominant contribution above $40 \times 10^{-3} m_0 c$. Extrapolating these results we can infer that the increase in the peak intensity ($10 < p_L < 25 \times 10^{-3} m_0 c$ range) in the transition elements of Fig. 3 can be correlated with the increase of the annihilation of positrons with 3d electrons: Ti, V, Cr,

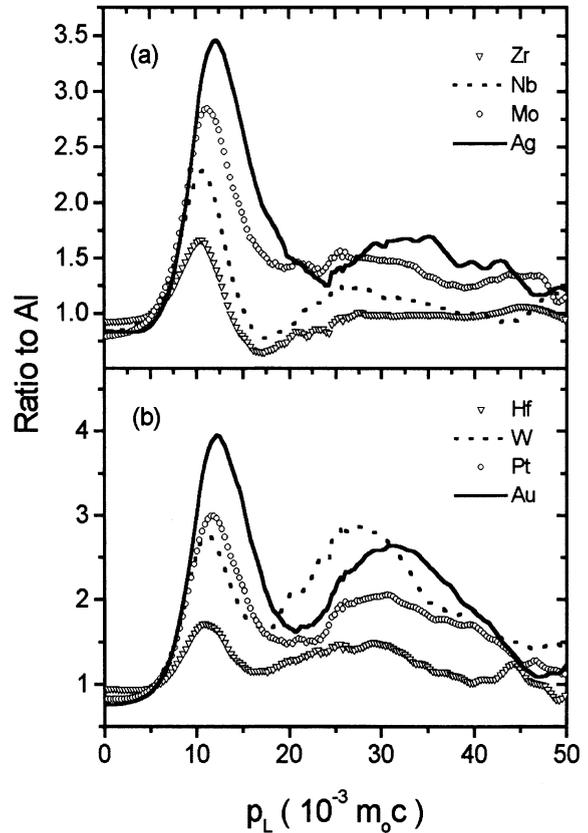


Fig. 4. Ratio of the experimental (a) Zr, Nb, Mo, Ag and (b) Hf, W, Pt, Au, positron–electron momentum distribution to the positron–electron momentum distribution of Al.

Mn, Fe, Co, Ni, Cu have 2, 3, 5, 5, 6, 7, 8, 10 electrons in the 3d orbital respectively. The broadening of the peaks from Fe to Cu is probably due to the increasing contribution from the more localized 3p electrons.

Cr, Mn, Fe, Co samples used in this investigation have shown the presence of positron traps even after annealing (see Table 1). The relation among the height of the peaks is preserved even if we account for the presence of these traps. The Zn curve does not follow the trend observed in the ratio curves from Ti to Cu: it is very broad and lower than the Cu curve. This is possibly due to the complete filling of the 3d and 4s orbitals.

The Ga (29.7°C melting temperature) ratio curve shows a high annihilation probability with valence electrons because of the presence of open

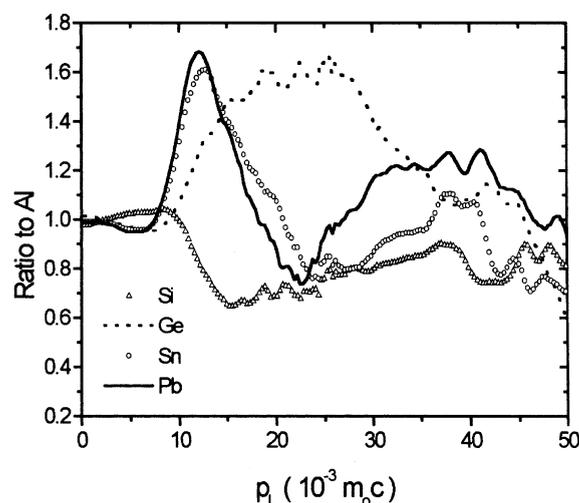


Fig. 5. Ratio of the experimental Si, Ge, Sn, Pb, positron-electron momentum distribution to the positron-electron momentum distribution of Al.

volume defects in our samples (see the lifetime value in Table 1). This curve is characterized by a small peak at $10.2 \times 10^{-3} m_0 c$.

Fig. 4(a) and (b) shows measurements on transition metals belonging to the fifth and sixth row of the Periodic Table. The ratio curves of these elements have similar features. Every ratio curve shows a first peak around $10\text{--}12 \times 10^{-3} m_0 c$ followed by a second broad peak in the $(20 < p_L < 40) \times 10^{-3} m_0 c$ region. The position of the first peak moves, in both the series, from low to high momenta (Fig. 6). The second broad structure is more pronounced in the Hf–Au series. The amplitudes of both peaks increase going from Zr to Ag and from Hf to Au. The main contribution to the two peaks in the first series is due to the 4d and 4p electrons [24]: the increase in the second peak height going from Zr to Ag is probably due to the filling of the more delocalized 4d orbital. In the case of the Hf–Au series the increase of the height of the two peaks can be correlated to the filling of the 5d orbitals [24].

In Fig. 5 we have grouped Si and Ge semiconductors that have a diamond-like structure and Sn and Pb metals which show a similar shape in the ratio curves. In Si and Ge there is a large open volume between atoms due to the diamond-like

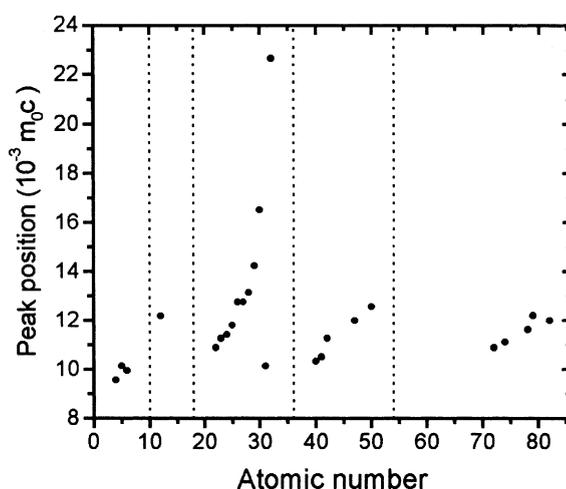


Fig. 6. Position of the first peak in the ratio between the positron-electron momentum distribution of the element to the positron-electron momentum distribution of Al, versus the atomic number Z .

structure. In these elements the positron wave function is localized in this open structure and the probability of core annihilation is small. In Si the core annihilation is mainly due to 2p electrons like in Al, but the predicted annihilation rate is smaller than in Al [8]. For these reasons the Si ratio curve is almost flat up to $10 \times 10^{-3} m_0 c$ and then drops below 1 in the region where the annihilation contribution from the 2p electrons is expected. In Ge the major contribution to core annihilation comes from positron annihilation with 3d electrons well localized in the r space [8]. This manifests in the broad peak centered around $22 \times 10^{-3} m_0 c$. The height of this peak is lower than the height of the peaks of transition metals in Fig. 3, which has been mainly attributed to annihilation with the 3d electrons. In Sn (metallic α phase), the most important part of the core annihilation has been calculated to be due to 4d electrons more localized in the momentum space [8]. This localization of the electrons in the momentum space lead to the more defined peak in Fig. 5. Extrapolating the results for Sn we expect that for Pb, that shows a ratio curve very similar to that of Sn, the annihilation comes mainly from 5d electrons (Sn and Pb have the 4d and 5d orbitals filled and 4 valence electrons $5s^2 p^2$ and $6s^2 p^2$, respectively). We note

that Pb and Sn, like Ga, have a low melting point, as a consequence the equilibrium concentration of thermal vacancies is expected at room temperature.

The data presented through Figs. 2–5 show as each element is characterized by features in the ratio curves: one or two peaks with different intensities and positions. The position of the first peak of all elements versus the atomic number Z is summarized in Fig. 6. This plot evidences four groups, each corresponding to elements in the same row in the Periodic Table. In each group the peak position starts from low momenta and moves toward higher momenta. The height of the first peak (Figs. 2–5) follows the same behavior except for Zn, Ge, Ga, Sn, Pb. The height and the position of the second peak, where detectable, seem to follow the same tendency as the first peak.

Although the height of the peaks has a discernible behaviour along the Periodic Table rows, its absolute value could be affected by positron annihilations in positron traps and this sets a limit to the present Doppler broadening measurements in pure elements. The height of the peak could be affected by: (a) positron annihilation with core electrons of impurity atoms (the purity of our samples is reported in Table 1) and/or annihilation in impurity–vacancy complexes; (b) positron annihilation into residual vacancies or vacancy agglomerates. The positron annihilations in such states, decreasing the annihilations with the core electrons of the element under study, reduce the intensity of the peaks in Figs. 2–5. This will be discussed in the following.

4.2. Pure elements: effects due to positron trapping

Annihilation of trapped positrons changes the probability of positron annihilation with core electrons and consequently the shape of the momentum spectrum. The high-momentum tail of the annihilation line is found to be very sensitive to a small perturbation caused by defects. In this paragraph we will present an experimental investigation of this effect on four samples.

In Fig. 7(a) and (b) the ratio curves for zirconium and niobium, before and after the thermal annealing are presented. One annealing only was

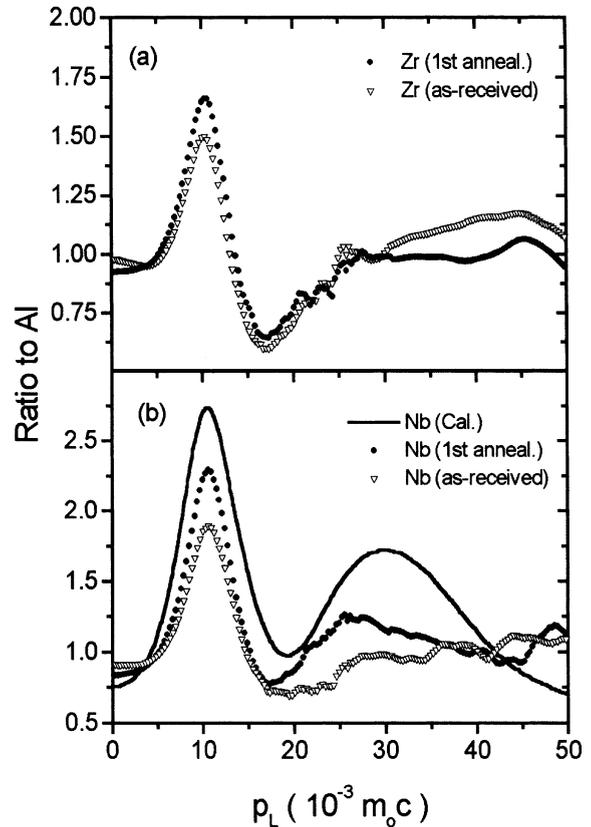


Fig. 7. Ratio of the experimental positron–electron momentum distribution to positron–electron momentum distribution of Al before and after annealing for (a) Zr (b) Nb. The continuous line is the calculated ratio curve from [24].

sufficient to eliminate the open volume defects from the as-received samples. Before annealing the decomposition of the lifetime spectrum for Zr samples gave two positron lifetimes: $\tau_1 = 154.4 \pm 2$ ps ($I_1 = 89.2 \pm 3\%$) and $\tau_2 = 269 \pm 18$ ps ($I_2 = 10.8 \pm 3\%$), average lifetime $\tau_a = 167$ ps. After annealing, a single lifetime of 159 ps was found in very good agreement with the free defect bulk lifetime (see last column of Table 1). The ratio curve for the annealed samples increases in the $5\text{--}30 \times 10^{-3} m_0 c$ region without changing the shape and this increase corresponds to the disappearance of small vacancy agglomerates.

In Nb before annealing only one lifetime was present (173 ps), indicating the presence of vacancy defects. After annealing this lifetime decreases to

127 ps (free defect bulk lifetime). In the Nb ratio curves, the second peak, due mainly to annihilations with 4d and 4p electrons [24], is partially suppressed by positron annihilation in trapping centers.

The ratio curves of Zr represent a typical behavior of the ratio curves in the majority of pure elements during the removing the open volume defects. An increase of the amplitude of the high-momentum part of the ratio spectra but without variation in the shape was also observed in other elements like Ni, Fe, Cu, Zn, Mo. In these elements the presence of defects influences the total annihilation probability with core electrons.

As a second example in Fig. 8(a) and (b) we report the ratio curves for W and Au. In W and

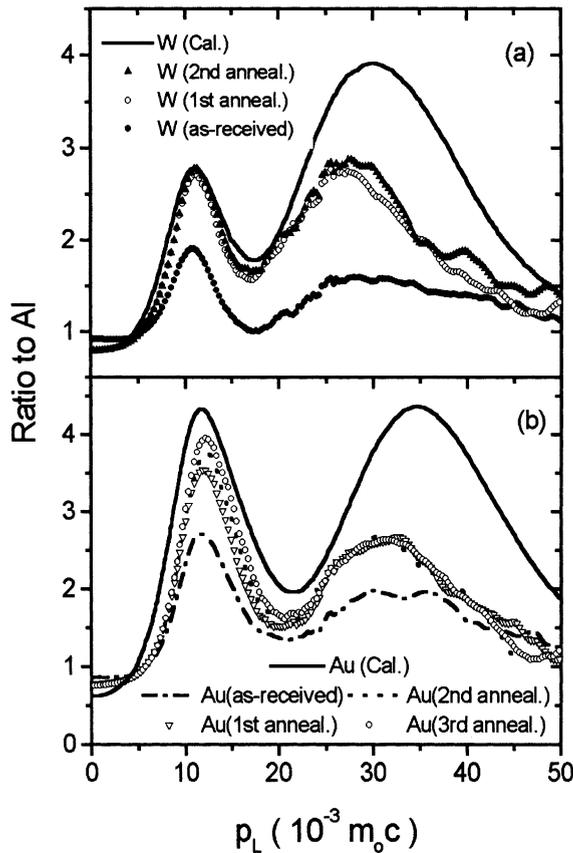


Fig. 8. Ratio of the experimental positron–electron momentum distribution to positron–electron momentum distribution of Al before and after annealing for (a) W (b) Au. The continuous lines are calculated ratio curves from [24].

Au more than one annealing was done but this was not sufficient to eliminate all the positron traps in our samples. Despite of this, the ratio curves for these elements are well suited to show different characteristic trends during the annealing procedure and are representative of the behavior of the ratio curves of other elements.

The lifetime spectra in the W as received samples, consist of a single lifetime of 175 ps, indicating that our starting sample is rich of small open volume defects [32]. The first annealing causes a big effect: the intensity of the two peaks increases. The second annealing produces a further small increase in the intensity of the peaks. In particular the second peak slightly surpasses in height the first peak. The lifetime spectra measured after the two annealing have been decomposed in two components. The values obtained for the lifetimes and their intensities were, within the errors, the same after the first and the second annealing (see Table 1 for the values). On the contrary, as can be seen from Fig. 8(a), the two ratio curves differ. In this case the ratio curves are more sensitive than the lifetime analysis to a small change in the defect concentration and/or defect type.

The first annealing in Au produces a big increase in the intensity of the two peaks. The starting material was rich of several positron traps ($\tau_1 = 181 \pm 3$, $I_1 = 92 \pm 5$, $\tau_2 = 291 \pm 37$, $I_2 = 8 \pm 5$). After the first annealing an increase of the amplitude of the two peaks is observed. Differently than in W case, the second and third annealing have major influence on the first peak, with no influence on the second one. After the first annealing τ_1 decreases to 118 ± 1 ps ($I_1 = 93 \pm 1$) and τ_2 to 256 ± 16 ps ($I_2 = 7 \pm 1$). Like in the W case, the second annealing has not produced a variation in the lifetime spectra, but we observe in Fig. 8(b) a big increase of the intensity of the first peak. The last annealing (see lifetime data in Table 1) is characterized by a change in the intensity of the two lifetime component. In the Au case it is evident the movement of the position of the first peak towards higher momenta, during the removal of open volume defects. This behavior in the peak position was found in other elements like Cr and Co.

4.3. Comparisons with other results

In Ref. [24], theoretical ratio-curves to Al were reported for Fe, Ni, Cu, Zn, Ge, Nb, Mo, Ag, Sn, W, Au, Pb. Experimental ratio curves to Al were reported for Cu, Ni, Fe, W, Ge, C, Sn, Si [9]. Our experimental results agree qualitatively both with the theoretical and the measured curves. But, before outlining the major similarities and differences, some points must be stressed. It has been shown in a case study on Cu [21] that the ratio curves change if HPGe detectors of different energy resolutions are used to acquire the momentum spectra. The general shape of the ratio curve remains the same by worsening the detector resolution but the ratio curve decreases in intensity and moves towards higher momenta. We have measured our samples with a detector resolution of 1.3 keV, consequently the peaks in our ratio curves can be expected to be slightly higher than the peaks in the ratio curves calculated convoluting the theoretical results with a gaussian with 1.5 keV FWHM to simulate the detector resolution [24] and higher than the measurements obtained with such detector resolution [9]. The model used to make the theoretical calculation reported in [24], as outlined by the authors, does not give a good description of the valence electrons and the calculated ratio curves can be only used for qualitative comparison below $20 \times 10^{-3} m_0 c$.

The theoretical ratio curves of [24] are higher than our experimental data but the general shape, the displacement of the first peak and the increase of its amplitude with the position of the elements in the periodic table are in a good agreement with our findings. For a comparison with our experimental data we have reported theoretical curves, from [24], for elements belonging to different rows in the periodic table. Ni and Pb ratio curves are shown in Fig. 9, Nb curve in Fig. 7 and W and Au curves in Fig. 8. Above $20 \times 10^{-3} m_0 c$ the shape of all our measured curves is in agreement with theory. The maximum of the second peak in Nb, W and Au is shifted toward lower momenta as expected from considerations on the detector resolution. The theoretical model, as said, is not precise below $p_L = 20 \times 10^{-3} m_0 c$, but we stress the similarity between theoretical

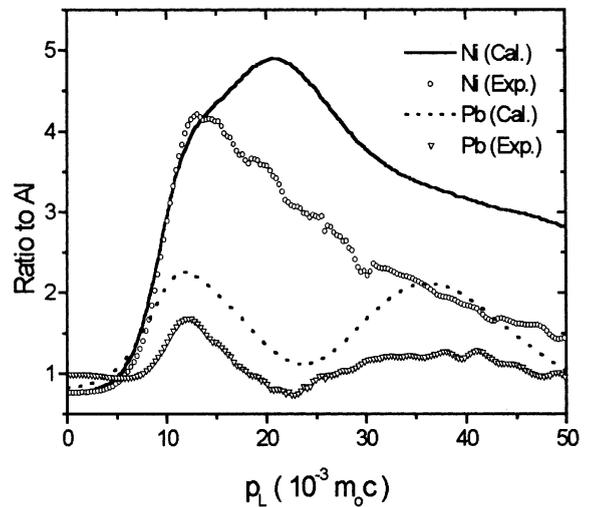


Fig. 9. Comparisons between theoretical (from [24]) and present experimental positron–electron momentum distribution ratio-curves with Al, for Ni and Pb.

and experimental curves also in this momentum region.

The overall agreement is found also with the measurements of [9]. Here, the position of the maxima are slightly displaced towards higher momenta as expected by the different detector resolution [21]. The peaks are generally higher than ours. The change in the height of the peaks is in the opposite direction than expected from considerations on detector resolution only. Similar differences were stressed between results obtained in [8,9] and can be attributed to the differences in the experimental set-ups: in [9] a two Ge detector system was used; in [8] a NaI(Tl) combined with a Ge detector was used as in the present measurement. Part of the difference in the height of the ratio curves, especially at high momenta ($p_L > 20 \times 10^{-3} m_0 c$), can be ascribed to a higher background in the NaI(Tl)-Ge system and probably to a different content of residual positron traps. Unfortunately in the previous papers published on this subject the purity of the measured samples is reported only in few cases. More reliable calculations below $20 \times 10^{-3} m_0 c$ and measurements on samples of different purity, will be useful for more accurate comparisons. We finally note that the FWHM of the peaks in the ratio curves is smaller

with a better detector resolution, which is the case of the present measurements.

5. Conclusions

Recent works [22,23,25] have shown that impurity elements give a precise signature in the Doppler broadened 511 keV annihilation line when positrons annihilate in impurity-defect complexes. This signature can be evidenced in momentum spectra measured by a two-detector coincidence technique. Here the position of the peaks is characteristic of the element impurities surrounding the annihilation site.

This paper provides the first large collection of measurements which can be used in the identification of elements and that constitute the starting point for constituting a future database. Elemental samples used in the measurement were of good purity and have been characterized by positron lifetime measurements. This work shows that the two-detector coincidence technique is very sensitive to the presence of positron traps in the samples and that it is more sensitive than the lifetime technique in the detection of positron traps. This finding has to be taken into account for precise comparisons among different sets of experimental data or between theory and experiment.

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